

Low p-type contact resistance using Mg-doped InGaN and InGaN/GaN superlattices

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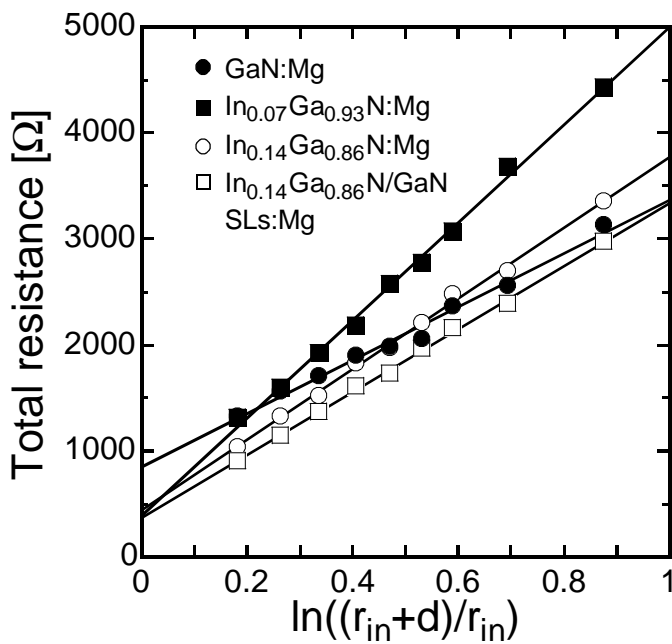
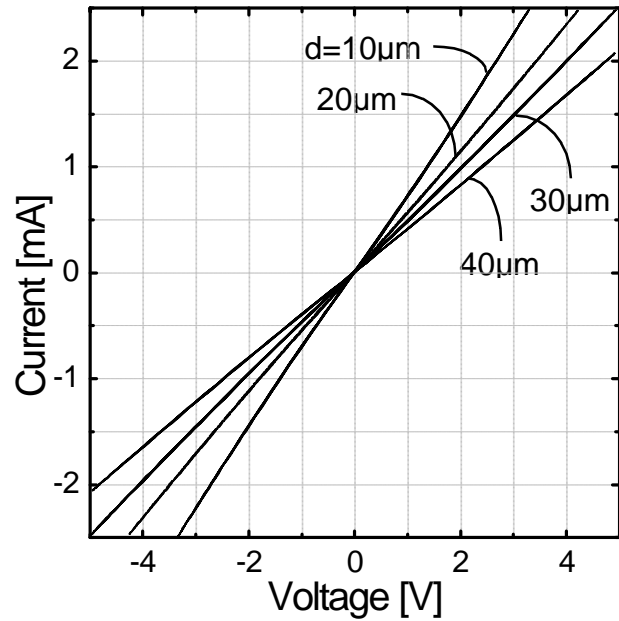
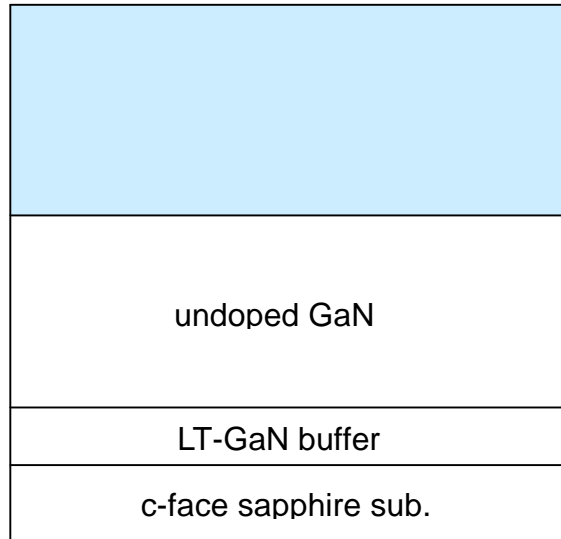
Recent significant progress in growth of group-III nitride-based semiconductors by metalorganic vapor phase epitaxy has made it possible to realize InGaN-based light emitting diodes and laser diodes. However, it is difficult to grow p-type GaN with high hole concentration due to relatively deep acceptor level and there are no appropriate metals having large work functions, resulting in high p-type contact resistances. To further improve the device performances, one of the important problems is to obtain the low-resistance ohmic contacts to p-type materials. To date, the metallization schemes and surface treatments strongly affected the contact resistances,¹⁾ resulting in the order or less than $10^{-4} \Omega\text{cm}^2$. From the viewpoints of high hole concentration layers, p-type AlGaIn/GaN superlattices (SLs) were used to obtain low contact resistance.²⁾ Recently, we realized p-type InGaIn with hole concentrations above 10^{18} cm^{-3} and p-type InGaIn/GaN SLs with those above 10^{19} cm^{-3} .^{3,4)} In the GaAs system, a heavily doped narrow band-gap InGaAs has been widely used as a cap layers, because InGaAs has a narrower band-gap energy (E_g) than GaAs, and an InGaAs cap layer reduces both the contact potential barrier and the contact resistance. Therefore, InGaIn having narrower band-gap than GaIn is expected to be used as the low-resistive contact layers of group-III nitride-based semiconductors. So far, for n-type contacts, InN/GaN short-period SLs have been demonstrated for the cap layers.⁵⁾ In this work, we investigated p-type InGaIn and InGaIn/GaN SLs as p-type ohmic contact layers.

We first grew a GaN buffer layer deposited at 500°C and a $1\text{-}\mu\text{m}$ -thick undoped GaN at 1000°C on c-face sapphire substrates. Then, we grew Mg doped GaN at 1000°C or Mg-doped $\text{In}_{0.14}\text{Ga}_{0.86}\text{N}$ or $\text{In}_{0.14}\text{Ga}_{0.86}\text{N}/\text{GaN}$ SLs at 780°C . The sample structure of Mg-doped InGaIn and SLs is schematically shown in Fig. 1. The source materials were triethylgallium, trimethylindium and NH_3 with N_2 as a carrier gas for the InGaIn growth. The p-type dopant source was bis-cyclopentadienylmagnesium. All the samples were annealed at 700°C in N_2 ambience after the growth. Ni/Au ($20 \text{ nm}/20 \text{ nm}$) metal was deposited using electron beam evaporation system. The hole concentrations of the Mg-doped GaN, $\text{In}_{0.07}\text{Ga}_{0.93}\text{N}$, $\text{In}_{0.14}\text{Ga}_{0.86}\text{N}$ and $\text{In}_{0.14}\text{Ga}_{0.86}\text{N}/\text{GaN}$ SLs were 6.9×10^{17} , 2.7×10^{18} , 6.7×10^{18} and $2.1 \times 10^{19} \text{ cm}^{-3}$, respectively.

The specific contact resistances(ρ_c) were characterized by current-voltage (I-V) measurements based on the circular transmission line model (CTLTM). The CTLTM pattern was designed with a constant inner radius ($r_{in}=50 \mu\text{m}$) and spacings (d) from 10 to 70 μm . Figure 2 shows the typical CTLTM I-V characteristics of the Mg-doped GaN. Linear I-V characteristics indicated that the good ohmic contacts were achieved in our process. Figure 3 shows the total resistance (R) as a function of $\ln((r_{in}+d)/r_{in})$. The data of R were determined by the CTLTM I-V characteristics as shown in Fig. 2. Since the I-V curves of all samples showed linear behavior, we could determine ρ_c from the data in Fig. 2. Table I shows ρ_c of the Mg-doped GaN, $\text{In}_x\text{Ga}_{1-x}\text{N}$ ($x=0.07, 0.14$) and $\text{In}_{0.14}\text{Ga}_{0.86}\text{N}/\text{GaN}$ SLs. While ρ_c of Mg-doped GaN is $1.1 \times 10^{-2} \Omega\text{cm}^2$, ρ_c of Mg-doped InGaIn and SLs are $1\sim 3 \times 10^{-3} \Omega\text{cm}^2$. These lower ρ_c of the Mg-doped InGaIn system are ascribed to narrower E_g and higher acceptor concentrations in InGaIn⁴⁾ and InGaIn/GaN SLs.

These results indicate that the Mg-doped InGaIn system is promising for lowering the p-type contact resistance. Further optimization of metallization schemes and surface treatment will improve ohmic contact characteristics, resulting in high device performance and reliability.

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- 4) K. Kumakura, T. Makimoto and N. Kobayashi, Jpn. J. Appl. Phys. **39** (2000) L337.
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Mg-doped samples	Specific contact resistance [Ωcm^2]
GaN	1.1×10^{-2}
$\text{In}_{0.07}\text{Ga}_{0.93}\text{N}$	1.4×10^{-3}
$\text{In}_{0.14}\text{Ga}_{0.86}\text{N}$	2.3×10^{-3}
$\text{In}_{0.14}\text{Ga}_{0.86}\text{N/GaN SLs}$	1.9×10^{-3}